

Proton microbeam irradiation effects on PtBA polymer

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MS received 13 November 2004

Abstract. Proton beam lithography has made it possible to make various types of 3D-structures in polymers. Usually PMMA, SU-8, PS polymers have been used as resist materials for lithographic purpose. Microbeam irradiation effects on poly-tert-butyl-acrylate (PtBA) polymer using 2.0 MeV proton microbeam are reported. Preliminary results on pattern formation on PtBA are carried out as a function of fluence. After writing the pattern, a thin layer of Ge is deposited. Distribution of Ge in pristine and ion beam patterned surface of PtBA polymer is studied using the optical and secondary electron microscopic experimental methods.

Keywords. Ion micro-beam; pattern generation; PtBA polymer; Ge on polymers.

1. Introduction

Recently, the fabrication of sensor and actuator devices on the micro-scale and their integration with electron devices and micro-electromechanical systems (MEMS) has gained a highly technological importance (Ikuta and Hirowatari 1993). With the advent of nanoscience, nano-electromechanical systems (NEMS) show promising future (Roukes 2001). The majority of sub-micron machining procedures involve two types of interactions with the resist materials: electromagnetic radiation (e.g. optical, UV or X-ray photons) or charged particles (electrons, low energy heavy ions, high energy light ions). In general, the micromachining procedures based on electromagnetic radiation require masks. The use of mask with the charged particles is not convenient, as the greater energy deposited in the mask during exposure results in mask instabilities due to heat expansion, stress and damage. Thus, the ion beam micromachining is limited to direct writing processes (Breese *et al* 1996). The photolithography utilizes optical light passing through a patterned mask. In the photolithography, the transmission of light exposes a specific geometrical pattern in a resist layer deposited on a substrate and depending on the positive or negative type of resist material, the exposed or unexposed part gets chemically etched out forming the microstructure. Although photolithography is essentially a surface micromachining technique and therefore two-dimensional,

various wet and dry etching techniques have been successful in producing 3-D microstructures. X-ray lithography has the capability of manufacturing 3-D structures with high aspect ratios because of the penetration properties of X-rays in resists such as polymethylmethacrylate (PMMA). When the developed resist is used in conjunction with electro-deposition, which fills the resist mold with metal, then 3-D metallic microstructures can be produced. This process is now universally known by its German acronym LIGA. The use of LIGA is limited for mass production as it requires higher photon flux using synchrotron radiation sources (Rogner *et al* 1992; Patenburg and Mohr 2001; Madou 2002).

Among the high energy charged particle lithography, proton beam micromachining (PBM) [or deep ion beam lithography (DIBL)] is a well known method (Breese *et al* 1996; Watt 1999; Gonin and Munnik 2003). PBM uses energetic proton beams (few MeV energy) for patterning in resist materials. It is advantageous to use PBM compared to other 3-D lithographic techniques as it is a direct writing process (i.e. no masks are required), control in construction of slots, channels, holes, complex non-prismatic shapes, etc (Watt 1999; Gonin and Munnik 2003). But the PBM process is slow for commercial direct high volume batch production of microcomponents. The use of quadrupole lenses to obtain a finely focussed beam spot coupled with the ability to scan the beam in complicated patterns has resulted in microstructures of intricate 3D shapes and smooth side walls (Watt 1999; Gonin and Munnik 2003).

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Study of ion induced modification in polymers has been an active area for last two decades (Hioki *et al* 1983; Rao and Lee 1996). A basic understanding of physical and chemical modifications enables the use of polymers in lithography, preparation of filters, window materials for external ion beam applications such as external particle induced X-ray emission (PIXE), etc. Ion irradiation of polymers can induce irreversible changes in their macroscopic properties such as electrical and optical properties (Hioki *et al* 1983) and mechanical properties (Rao and Lee 1996). Electronic excitation, ionization, chain scission and cross-links as well as mass losses are accepted as the fundamental events that led to the observed macroscopic changes (Abel *et al* 1995). It is known that high energy ion-induced effects result in track formation in polymer (Bouffard *et al* 1995). Recently, various groups at the Nuclear Science Centre, Delhi, had been working on various aspects of modification in polymers, like track formation, hydrogen loss, etc using swift heavy ions (Avasthi *et al* 1998; Rao *et al* 2003; Mujahid *et al* 2004). Zhu *et al* (2002) concluded that not only the physical processes but also the chemical processes of the energy deposition determine the modification of polymer.

For lithographic purpose, various types of resist materials have been used. Among them, PMMA and SU-8 were successfully used for PBM. While PMMA is a positive type of resist, the SU-8 is a negative type. Su-8 requires relatively less dose but the aspect ratios obtained are smaller compared to those obtained using PMMA (van Kan *et al* 1999). In this article, we present preliminary results using Institute of Physics microbeam facility for the purpose of patterning with a radiation damage on PtBA polymer. Poly(tert-butyl acrylate) (PtBA) is an amorphous homopolymer. This polymer is found to be resistant to photo-oxidation and has a glass transition temperature, $\approx 49^\circ\text{C}$ (Ahn and Shull 1996; Richter *et al* 2001).

With the impact of progress in nanoscience, active research has started on the 'bottom-up' approach. The 'bottom-up' approach could enable to build nanodevices with an atom or molecule as a base unit. Feynman gave a visionary talk in 1959 titled 'there is plenty of room at the bottom' in which he essentially argued for 'bottom-up' approach (http). The other approach, 'top-down' has been the basis for integrated circuit technology in the past. The two approaches, the 'top-down' and the 'bottom-up' needed to be efficiently combined to realize nanostructures for technological applications. Growth of various semiconducting and metal nanoparticles on a polymer follow the 'bottom-up' approach as the self-assembly dominates. But in this type of self-assembly, in general, in all self-assembly processes, there is no control over the position or placement. To control the placement of the nanoparticles following self-assembly, one requires a sort of pre-patterning. Das *et al* (2000, 2001) proposed the possibility of growth of one or two-dimensional lattice of Ge islands if the polymer surface is patterned. Zhong *et al* (2003, 2004) have

grown a two-dimensional periodic positioning of self-assembled Ge islands on pre-patterned Si(001) substrates. In order to understand the growth of semiconductor particles on a patterned surface of a polymer, proton ion microbeam for patterning the polymer surface and high vacuum deposition using resistive heating has been used.

Here, we report the first use of ion-microbeam facility at the Institute of Physics, for writing a pattern on PtBA polymer and growth of Ge islands on the patterned polymer surface.

2. Experimental

An ion microbeam facility has been set up at one of the beam lines of the 3 MV tandem Pelletron accelerator at the Institute of Physics, Bhubaneswar, in collaboration with the University at Albany, USA (Rout 2001; Rout *et al* 2001). A spatial resolution of $\approx 3 \mu\text{m}$ with a beam current of about 80 pA with a 3 MeV He^+ beam was achieved. Figure 1 schematically shows the microbeam facility. A set of micropolished slits and a powerful magnetic quadrupole doublet lens are used. The electrically isolated pole pieces of the lens are used as deflection plates to scan the beam across the sample. A motorized X–Y stage is used to position the sample. The target chamber houses three detectors: a cooled Si(Li) detector used to analyse X-rays, a surface barrier detector to analyse backscattered ions and an aluminized plastic scintillator for detecting secondary electrons from the target. A high resolution CCTV camera-microscope along with light guiding system helps to view the target optically. The system is equipped with a rear microscope for directly looking at the beam spot. The necessary hardware and software have been developed for control of the beam, data acquisition and auto-pumping/venting of the target chamber. Both RBS and PIXE data are collected, stored and displayed simultaneously through the attached computer. The same computer is also used to store the digitized image of the sample in the SEM mode and to control the X–Y stage for the sample holder. A representative pattern 'IOP' is depicted in the inset of figure 1. The position coordinates of 'IOP' is written in a file with 4096×4096 pixels depth, which eventually gets converted to respective voltages in X and Y scanners by the program for 2-D beam scanning. To scan the beam in a desired pattern one needs to give the set of X, Y coordinates to the program which then converts to proportional analog voltages and supplies to the X- and Y-deflectors to deflect the beam in a programmed fashion. The pattern is generated by writing X, Y, and Z coordinates in a text which is read by the program. The Z-coordinate corresponds to time for which the beam will stay at a given X, Y coordinate. At present, two modes of operation are available: continuous mode and point mode. In continuous mode, the beam position can be varied continuously. In point mode the beam stays at a position for

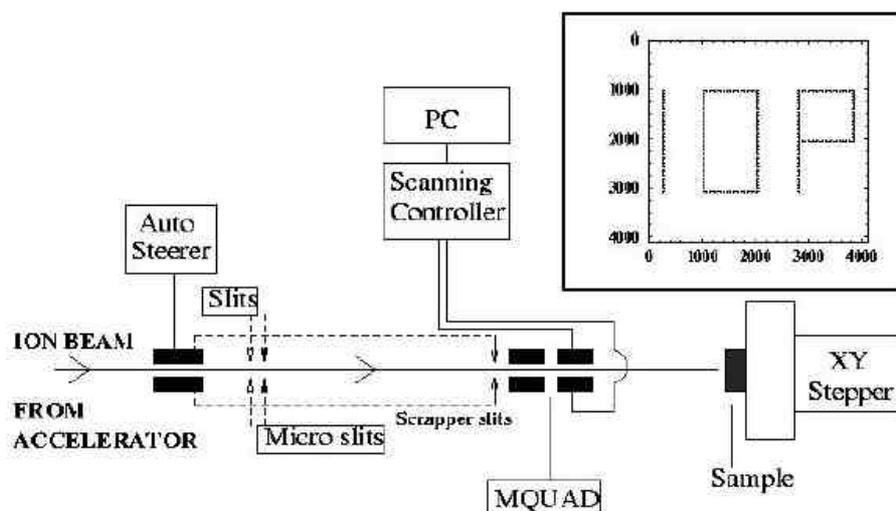


Figure 1. Schematic set up of ion-microbeam facility at the Institute of Physics and inset shows the position coordinates that are used to write 'IOP'.

a pre-set time and then moves to the next point. This will make the program more versatile as one can pattern in Z coordinate (which corresponds to more damage) also depending on the time at different points. For the present work we used the beam in continuous mode. Thus to begin with we wrote the X, Y coordinates in 4096×4096 pixels for 'IOP' which is shown in the inset of figure 1. Present writing program takes the top left corner as the origin.

Poly(tert-butyl acrylate) (PtBA) with a molecular weight of 3,01,000 was synthesized by anionic polymerization at -78°C in tetrahydrofuran as described in Ahn and Shull (1996). The PtBA thin films were prepared by spin coating using solutions of PtBA in butanol. The silicon substrates were cleaned prior to deposition. A 100 nm thick PtBA film was then coated at 2000 rpm. A 2 MeV proton beam was used for pattern writing. This beam can scan $250 \times 250 \mu\text{m}^2$ area with the existing set up. As the purpose of the present study is to make a pattern (radiation damaged area) and then deposit Ge islands on the patterned surface, a beam size of $\sim 10 \mu\text{m}$ was used. A direct measurement of total fluence was not possible and hence fluence has been controlled with the scan time. Scanning electron microscopic measurements were carried out at TIFR, Mumbai.

3. Results and discussion

MeV ions passing through polymer films modify their electrical and optical properties and these changes are related to changes in the chemical structures of the polymer. The penetration depth of proton beam in polymers, in general, is much more than the thickness of the PtBA film (e.g. 2 MeV proton penetrates $62 \mu\text{m}$ into PMMA). Figures 2 and 3 show optical micrographs after irradiation.



Figure 2. Optical micrograph of 'IOP' patterned on PtBA.

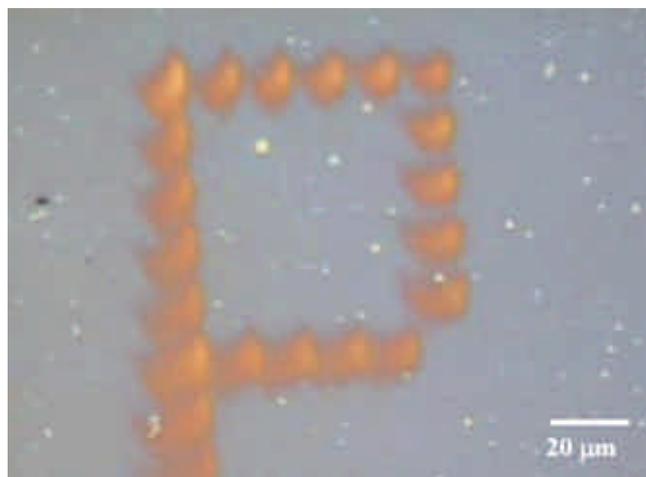


Figure 3. Optical micrograph of 'P' patterned on PtBA for effective longer exposure.

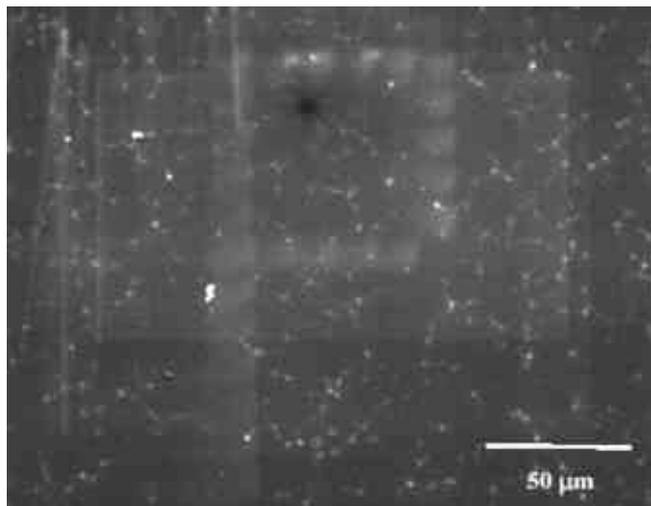


Figure 4. SEM micrograph of Ge (≈ 10 nm thick) deposition on PtBA surface after ion beam writing.

tion. The ion beam typically of $10\ \mu\text{m}$ size has been programmed to scan in the pattern 'IOP' for about 2.5 h. We could not measure the current of the beam and also the total dose. At constant source current of $\approx 1.0\ \mu$ (typically, this results in ≈ 50 pA at the target in $\approx 3\ \mu\text{m}$ spot size), irradiation time was varied to change the total fluence on the sample. The optical micrograph in figure 2 shows the pattern 'IOP' with contrast in colour. A scan for 2.5 h irradiation time is used for this purpose. The optical properties change and hence the ion-irradiated location has a different colour. From figure 2, it is evident that our facility can be used in principle for microbeam patterning applications. A scan for 0.5 h irradiation time under similar conditions showed no change in colour at the irradiated part of the polymer. It appears that a threshold fluence might be present for change in colour to be observed. To have larger fluence, for total 2.5 h of irradiation, writing has been carried out only for 'P' instead of 'IOP'. Figure 3 shows an optical image of 'P'. The contrast for 'P' is better than that of 'IOP' due to more time of exposure. Our next step is to look for any preference in the growth of Ge on the patterned polymer surface.

Ge nanoparticles have been grown usually on silicon substrates by molecular-beam epitaxy (MBE) and by other methods. The lattice mismatch between Ge and Si and the consequent strain leads to a layer-plus-island growth process (known as Stranski–Krastanow (SK) grow mode) in which Ge islands grow on a uniform Ge layer. In the SK mode growth, it is not possible to grow isolated Ge quantum dots. Using other methods, isolated self-assembled Ge nanostructures have been grown on polymers by thermal evaporation under high vacuum conditions utilizing the non-wetting condition given by the surface energy relation: $S_{\text{Ge}} \gg S_{\text{polymer}}$ (Das et al 2000, 2001). In these studies, self-assembled nanostructural islands, isolated or coalesced on the line defects into lattice-like or continuous

structure have been observed to form at nominal film thickness (≈ 5 nm) of Ge on 35 nm thick polymer blend (15% polystyrene (PS), 85% poly(*p*-bromostyrene-styrene) ($\text{PBr}_{0.06}\text{S}$)) (Das et al 2000, 2001). Growth pattern varied from isolated islands of Ge to wires of Ge as the Ge film thickness was increased on PtBA polymer surfaces. Isolated Ge islands were found for a Ge thickness of ≤ 0.9 nm on the PtBA polymer surface (Satyam et al 2005). The change of growth pattern on the irradiated portion of the specimen will be an interesting issue. After irradiating in a programmed manner ('P' surface), a 10 nm thick Ge is deposited on the patterned PtBA surface. Figure 4 shows a SEM micrograph following Ge deposition. The energy dispersive X-ray spectrometry data (EDS) do not show any preferential growth of Ge in and around the irradiated part. Though the preferential growth of Ge is absent, it is required to do more precise measurements with nano-sized beams for patterning and smaller thickness of Ge on the nano-patterns and better imaging is necessary to understand this particular problem.

4. Conclusions

We have demonstrated the first use and capability of pattern writing using the ion microbeam facility at the Institute of Physics, Bhubaneswar. The patterned damage on PtBA polymer caused due to 2.0 MeV proton beams has been used to study growth of Ge nanoislands on pre-patterned polymer surfaces.

Acknowledgements

We would like to thank the staff of IBL, Institute of Physics, for their help for carrying out this experiment. Thanks to Ken Shull at Northwestern University for providing the polymer.

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