

Non-linear I – V characteristics of doped $\text{Sn}_{1-x}\text{Ti}_x\text{O}_2$ ($0.0 < x < 0.25$) system

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Abstract. The solid solutions of $\text{Sn}_{1-x}\text{Ti}_x\text{O}_2$ ($0.0 < x < 0.25$) were prepared by the usual high temperature ceramic processing techniques with small amounts of dopants such as Nb, Co, Al. The electrical measurements ($\log I$ – $\log V$) on the ohmically metal electroded sintered pellets showed the non-linear behaviour. The non-linear coefficient was found to decrease from 12 to 3 with increase in x . On the other hand, the breakdown voltage (E_b) showed the increase with the increase in x . This complex electrical behaviour is explained on the effects of the three dopants (as varistor former, performance enhancer and highlighter) on the microstructural features of the dense pellet.

Keywords. Varistors; SnO_2 ; TiO_2 ; nonlinear coefficient; breakdown voltage.

1. Introduction

Voltage dependent resistors (varistors) are used to protect the electrical/electronic circuits from transient voltage surges (Matsuoka 1971; Einzinger 1987; Gupta 1990; Clarke 1999). The most commonly used varistors are ZnO-based and their non-linear electrical properties were well studied and documented (Clarke 1999). An important characteristic of ZnO-based varistors is their high breakdown voltage (3 V per grain boundary) which precludes them for low voltage applications. The search for low voltage varistors, i.e. lower E_b with high a values, are based on following guidelines: The material must be a doped high bandgap polycrystalline semiconductor. Further, as in ZnO ceramics two types of dopants are necessary: one that is soluble in the grains to affect the grain resistivity and the other a large insoluble ion that segregates at the grain boundaries. The search has resulted in the discovery of nonlinear I – V characteristics in SrTiO_3 (Fujimoto *et al* 1985), TiO_2 (Yan and Rhodes 1982) and SnO_2 (Santhosh *et al* 1997) based materials. For example, in the case of SnO_2 -based varistor, Nb and Co were used as varistor formers and Al acts as a performance enhancer. The various important parameters of the above varistors are given in table 1.

Although the dopants used were different in TiO_2 and SnO_2 , both show nonlinear I – V characteristics. Both crystallize in rutile (tetragonal) structure. Further, SnO_2 has a higher breakdown voltage ($E_b = 700$ V/mm) value whereas TiO_2 has a lower E_b (200 V/mm). Hence we have attempted to prepare a series of solid solution, $\text{Sn}_{1-x}\text{Ti}_x\text{O}_2$, with common additives to get an optimum E_b value. We

report here the results obtained from the investigations carried out on doped SnO_2 – TiO_2 system.

2. Experimental

The solid solution $\text{Sn}_{1-x}\text{Ti}_x\text{O}_2$ containing additives was prepared by the standard ceramic technique. The SnO_2 , TiO_2 along with dopants such as $\text{Al}_2(\text{SO}_4)_3 \cdot 16 \text{H}_2\text{O}$, CoO and Nb_2O_5 in the stoichiometric ratio were mixed and ground well and calcined at 1473 K for 24 h. The calcined powders were again ground and fired at 1473 K for another 24 h. The compositional formula for the series is $\text{Co}_{0.01}\text{Nb}_{0.0005}\text{Al}_{0.0005}\text{Sn}_{0.989-x}\text{Ti}_x\text{O}_2$. The calcined powders were mixed with a binder (polyvinyl alcohol 2 wt%) and pelletized (12 mm dia, 1 mm thick) at 2–3 tons. The pellets were sintered at 1573 K for 2 h. The sintered pellets were polished and silver contacts were obtained by curing Ag-paste and annealed at 873 K for 30 min. The structure related phase determination was studied by Philip 1730 X-ray diffractometer. The change in lattice parameters in the solid solution was also recorded using slow scan. The microstructure of the sintered pellets was observed using a Leica Cambridge 440 microscope. The I – V characteristics were measured by using Keithley electrometer (610 C) and APLab (761 and 7323) power supply.

The current–voltage relation of a varistor is given by the equation,

$$J = (E/C)^a, \quad (1)$$

where J is the current density, E the applied field, C the proportionality constant and a the nonlinear coefficient. The current–voltage curves were plotted on log–log scale from which the slope of the curve gives the value of a .

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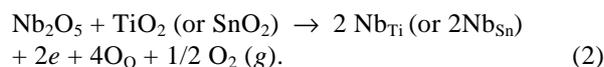
The another important parameter, E_b breakdown voltage is taken as the field applied when current flowing through the varistor is 1 mA.

3. Results and discussion

The variation of the unit cell parameters of doped $\text{Sn}_{1-x}\text{Ti}_x\text{O}_2$ system is shown in figure 1. Both the cell parameters of the tetragonal rutile structure decreases with increase in TiO_2 content. The 'a' values decrease from 4.738 Å to 4.658 Å and 'c' decreases from 3.188 Å to 3.108 Å as x varies from 0.0 to 0.25. This is in accordance with decrease in the ionic radius as Sn^{4+} (0.71 Å) is substituted by Ti^{4+} (0.68 Å). When $x > 0.25$, rutile TiO_2 segregates as a second phase as revealed by XRD. The dopants being added in small quantities do not show up as second phase in XRD.

Figure 2 illustrates the J vs E plot for the compositions $\text{Co}_{0.01}\text{Nb}_{0.0005}\text{Al}_{0.0005}\text{Sn}_{0.989-x}\text{Ti}_x\text{O}_2$ ($0.0 < x < 0.25$). The nonlinear coefficient decreases with increase in Ti content whereas the breakdown voltage increases with Ti. The low current in the pre-breakdown region implies that the grain boundary resistance is high for the solid solutions. However, the grain boundary barrier or energy barrier is not effective as shown by low α values. The dopants used for the end member SnO_2 to improve the grain boundary barrier were not ideal for the solid solutions. The dopant

Nb acts as a donor both in SnO_2 or TiO_2 . The Nb^{5+} added in small quantities dissolves in the lattice, replaces Ti^{4+} or Sn^{4+} to give an extra electron to the conduction process as



Thus Nb^{5+} ions are also expected to increase the conductivity in the solid solution. The addition of Co^{2+} ions was mainly responsible for the densification of SnO_2 and in the presence of TiO_2 may form oxygen ion vacancies due to its lower valency. The third dopant, Al^{3+} ions, added as performance enhancer (i.e. to increase the α value in SnO_2), is not effective in the case of $\text{Sn}_{1-x}\text{Ti}_x\text{O}_2$ as they exhibit low α values. The variation of the nonlinear coefficient (α) and the breakdown voltage (E_b) with TiO_2 incorporation were depicted in figure 3. The α decreases whereas E_b increases with increase in x .

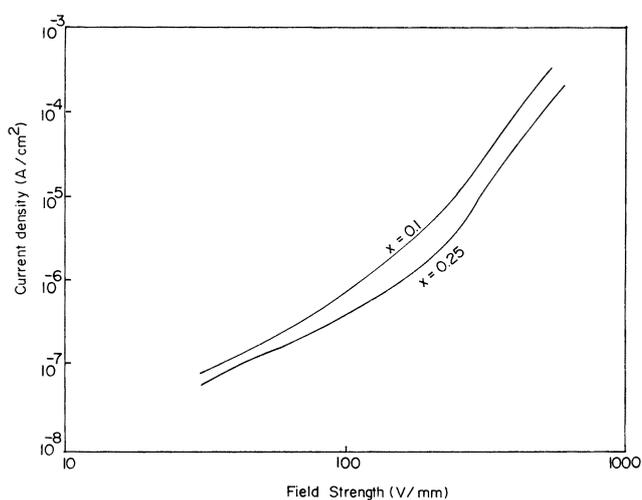


Figure 2. I - V characteristics of $\text{Co}_{0.01}\text{Nb}_{0.0005}\text{Al}_{0.0005}\text{Sn}_{0.989-x}\text{Ti}_x\text{O}_2$.

Table 1. Various varistor systems.

System	Dopants	α	E_b (V/mm)	Reference
SrTiO_3	Nb, Na, Si	20	100	Fujimoto <i>et al</i> (1985)
TiO_2	Nb, Ba	7	200	Yan and Rhodes (1982)
SnO_2	Nb, Co, Al	15	700	Santhosh <i>et al</i> (1997)

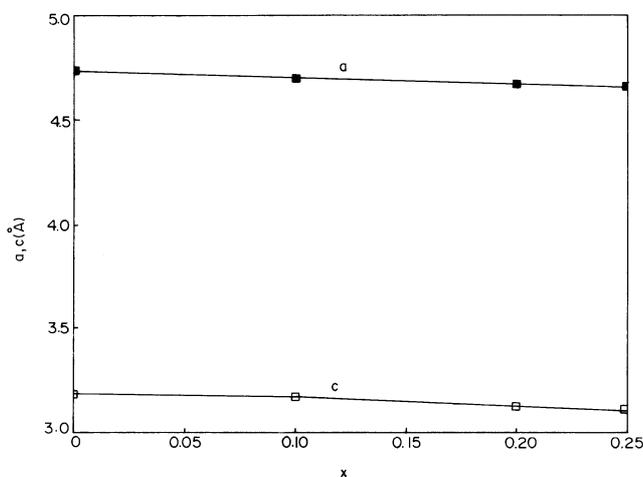


Figure 1. The variation of lattice parameters with x in $\text{Co}_{0.01}\text{Nb}_{0.0005}\text{Al}_{0.0005}\text{Sn}_{0.989-x}\text{Ti}_x\text{O}_2$.

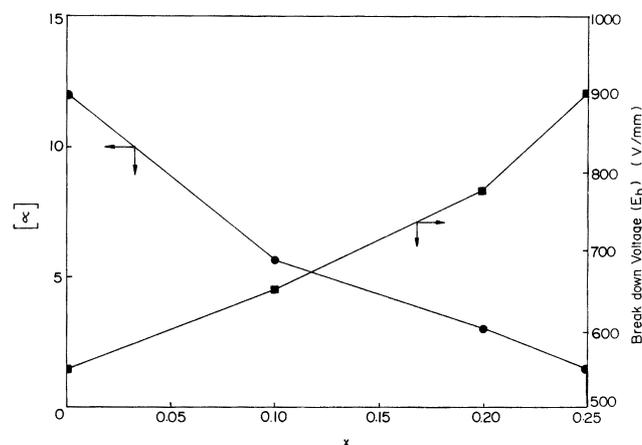


Figure 3. The variation of nonlinear coefficient (α) and breakdown voltage (E_b) with x in $\text{Co}_{0.01}\text{Nb}_{0.0005}\text{Al}_{0.0005}\text{Sn}_{0.989-x}\text{Ti}_x\text{O}_2$.

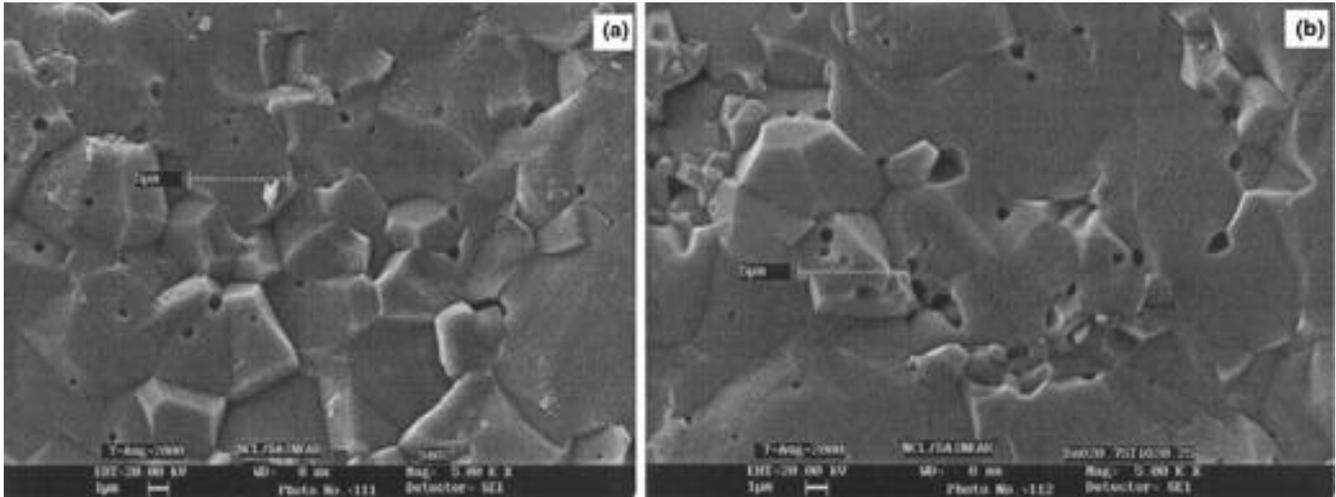


Figure 4. The microstructure of $\text{Co}_{0.01}\text{Nb}_{0.0005}\text{Al}_{0.0005}\text{Sn}_{0.989-x}\text{Ti}_x\text{O}_2$ for (a) $x = 0.0$ and (b) $x = 0.25$.

Figure 4 shows the microstructure recorded for the compositions $x = 0$ and 0.25 , sintered in identical conditions. The morphological features were well defined for the end member $x = 0$, whereas grain boundaries and triple points were not sharply defined for $x = 0.25$. This may be the possible reason for the low α values observed for the solid solutions. The average grain size was found to be 4μ in both cases. The substitution of Ti^{4+} for Sn^{4+} does not affect the grain growth characteristics as observed from micrographs.

4. Conclusions

The influence of isovalent ion such as Ti^{4+} on the I - V curves of doped SnO_2 was investigated in detail. The solubility of Ti^{4+} in SnO_2 is up to 25 mole%. The results show that α decreases for the solid solution whereas E_b increases with Ti^{4+} content. Presently systematic investi-

gations are on for other compositions to obtain optimum nonlinear coefficient and breakdown voltage.

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