

Some electrical and magnetic properties of γ -Fe₂O₃

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Abstract. γ -Fe₂O₃ synthesized from FeC₄H₄O₄·4H₂O has been studied using various techniques. The phase transformation observed by electrical conductivity measurements agrees well with the initial magnetization measurement. The magnetic hysteresis values compare with those of γ -Fe₂O₃ samples synthesized using established procedures. γ -Fe₂O₃ particles obtained were circular in shape showing a well resolved six narrow bands in Mössbauer spectrum. The presence of hydrogen ferrite phase was also confirmed by electrical and magnetic measurements.

Keywords. Electrical conductivity; Seebeck coefficient; ferrites; initial magnetization.

1. Introduction

γ -Fe₂O₃ is an important magnetic tape recording material and its synthesis has been studied in some detail (Kuo *et al* 1982; Karmazsin *et al* 1983; Ravindranathan and Patil 1986). However, studies on newer methods of synthesis and doping of γ -Fe₂O₃ to improve its magnetic properties are still being actively pursued (Rane *et al* 1981; Sharrock and Bonder 1985; White 1985; Venkataraman *et al* 1987a, b; Venkataraman *et al* 1989) because of its ideal combination of such magnetic hysteresis parameters as saturation magnetization (M_s), coercive force (H_c) and squareness ratio (M_R/M_S) (Bate 1975). Pure γ -Fe₂O₃ is a vacancy-ordered ferrite with a structural formula Fe₈³⁺[Fe³⁺ 40/3 □_{1/8}]O₃₂ where □ indicates vacancy sites present in the lattice.

The occurrence of hydrogen ferrite phase with the structural formula Fe₈³⁺[Fe₁₂³⁺H₄⁺]O₃₂ has been investigated by Rane *et al* (1981) and Venkataraman and Mukhedkar (1988) while synthesizing γ -Fe₂O₃ by thermal decomposition of ferrous oxalate dihydrate and ferrous fumarate half hydrate respectively. The magnetic properties of ferrites in general are influenced by the method of preparation, microstructure and crystal morphology (Schnettler 1972; Bate 1975; Manjula 1990; Vishwanadhan 1990). This paper reports studies on the magnetic behaviour of γ -Fe₂O₃ obtained by thermal decomposition of ferrous succinate tetrahydrate (FeC₄H₄O₄·4H₂O), and its suitability as a potent magnetic tape recording material. DC electrical conductivity was studied to understand the irreversible phase transformation (γ -Fe₂O₃ → α -Fe₂O₃) temperature and the result obtained is supplemented with Seebeck coefficient measurement and magnetic studies. γ -Fe₂O₃ and other ferrites, apart from their many technological uses, are also useful as potent heterogeneous catalysts.

2. Experimental

The experimental procedure for measuring the d.c. electrical conductivity, Seebeck coefficient, initial magnetization with variation of temperature, high field magnetic hysteresis loop tracer (HLT) and Mössbauer parameters were reported earlier (Rane *et al* 1981).

2.1 Synthesis

$\gamma\text{-Fe}_2\text{O}_3$ was synthesized using thermal decomposition of $\text{FeC}_4\text{H}_4\text{O}_4 \cdot 4\text{H}_2\text{O}$ under a controlled atmosphere of dynamic air containing water vapour at 290°C (Venkataraman 1987).

3. Results and discussion

Figure 1 shows the plot of $\log \sigma$ vs $1/T$ for $\gamma\text{-Fe}_2\text{O}_3$ under static air atmosphere. After the observed maximum V , there was a decrease in conductivity σ up to 130°C as indicated by region B' . The σ value then increased showing a Kink. With further increase in temperature, the σ value rose till a phase transformation of $\gamma\text{-Fe}_2\text{O}_3$ to $\alpha\text{-Fe}_2\text{O}_3$ was attained with broadening in the curve from the starting temperature T_D to the ending temperature T_E ($T_D - T_E$). The cooling curve showed hysteresis in which σ did not exhibit any variation between 180°C and room temperature (30°C). The occurrence of V was due to the desorption of the adsorbed water molecules and the Kink was due to existence of hydrogen ferrite phase. It is also seen from figure 1 that hydrogen ferrite phase was present up to 190°C at which the hydrogen ferrite decomposed to give a vacancy-ordered $\gamma\text{-Fe}_2\text{O}_3$ i.e. $\text{Fe}_8^{3+}[\text{Fe}_{12}^{3+}\text{H}_4^+]\text{O}_{32}$ to $\text{Fe}_{8/3}^{3+}[\text{Fe}^{3+} 40/3 \square_{1/8}]\text{O}_{32}$. This transformation is confirmed by magnetic studies discussed later. On further raising the temperature from 190°C , there was a slow transformation of $\gamma\text{-Fe}_2\text{O}_3$ to $\alpha\text{-Fe}_2\text{O}_3$ as indicated by temperature regions $T_D - T_E$ (440° to 530°C).

The Seebeck coefficient measurements (figure 2) showed a fall in the number of

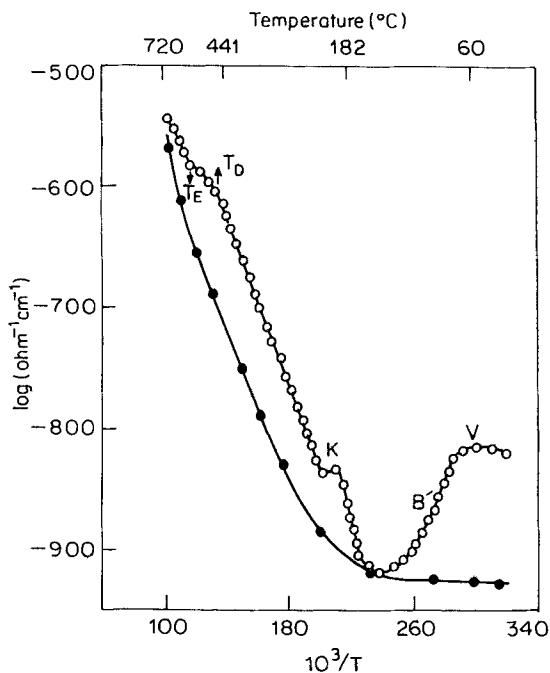


Figure 1. Plot of $\log \sigma$ vs T^{-1} (K) for $\gamma\text{-Fe}_2\text{O}_3$ heating curve; X, cooling curve.

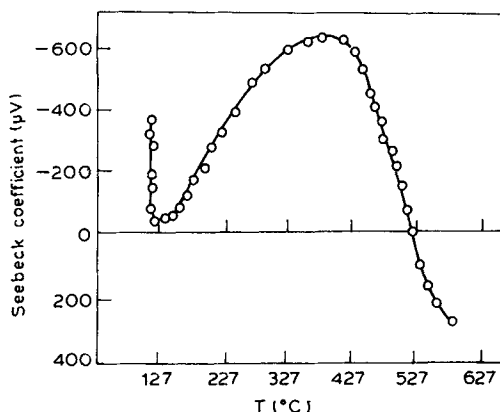


Figure 2. Plot of Seebeck voltage (μVK^{-1}) vs T^{-1} ($^{\circ}\text{C}$) for $\gamma\text{-Fe}_2\text{O}_3$.

Table 1. Magnetic parameters.

Sample	Saturation magnetization (M_s emu g^{-1})	Coercive force H_C (Oe)	Squareness ratio (M_R/M_S)
Hydrogen ferrite phase	65.0	250.0	0.53
Vacancy ordered $\gamma\text{-Fe}_2\text{O}_3$	72.0	350.0	0.58

charge carriers (negative type) in the temperature region 115–145 $^{\circ}\text{C}$ and then remained almost constant up to 190 $^{\circ}\text{C}$. A further increase in temperature showed a rise in the number of charge carriers up to phase transformation ($\gamma \rightarrow \alpha$) temperature, after which the nature of charge carriers changed from negative to positive type.

The magnetic hysteresis parameters for hydrogen ferrite and that of vacancy ordered $\gamma\text{-Fe}_2\text{O}_3$ (obtained by heating hydrogen ferrite under dynamic nitrogen atmosphere at 200 $^{\circ}\text{C}$ for 4 h) are given in table 1. The M_s value for vacancy ordered $\gamma\text{-Fe}_2\text{O}_3$ was higher than that observed for hydrogen ferrite. Such behaviour is expected when non-magnetic dopants are present in the crystal lattice (Venkataraman and Mukhedkar 1988). As expected all the magnetic parameters observed compare well with expected values (Bate 1975).

It was believed that permeability K or susceptibility X_i of a magnetic material having SD grains always increased on heating and showed a peak (Hopkinson's peak) just before T_C and then fell to zero rapidly (Bate 1975; Nagarajan and Murthy 1981). Figure 3 shows the normalized $X_i - T$ curve obtained for vacancy ordered $\gamma\text{-Fe}_2\text{O}_3$. The normalized X_i values varied with respect to temperature before Hopkinson's peak indicating SD behaviour of the sample, after this the X_i value decreased to zero T_C with some tailing.

The Mössbauer spectra of $\gamma\text{-Fe}_2\text{O}_3$ and that of hydrogen ferrite were found very similar with a well-resolved six narrow lines (half bandwidth of 0.285 mms^{-1}) in the intensity ratio 3:2:1:1:2:3. This spectra suggested that the effective magnetic fields at A and B sites of $\gamma\text{-Fe}_2\text{O}_3$ were comparable and the magnetic SD grains were randomly oriented (Venkataraman 1987). The hyperfine field for vacancy-ordered $\gamma\text{-Fe}_2\text{O}_3$ and hydrogen ferrite phase was found to be 493.8 KOe. Hence it can be

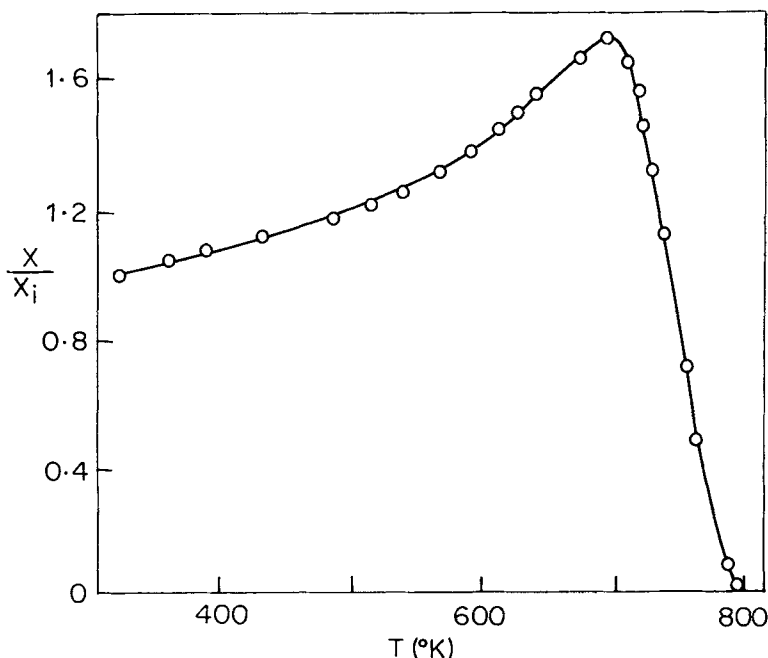


Figure 3. Plot of normalized susceptibility (X/X_i) vs $T(^{\circ}C)$ for $\gamma\text{-Fe}_2\text{O}_3$.

stated that the presence of nonmagnetic dopant like H^+ leading to the formation of hydrogen ferrite does not affect the magnetic fields at A and B sites. $\gamma\text{-Fe}_2\text{O}_3$ thus obtained showed spherical particles bound together to give particles of 1–6 nm diam. (Venkataraman 1987).

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